

Effects of the Ar ions pre-amorphization of Si substrate on interface mixing of Fe/Si bilayers

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Summary

Ion beam mixing of Fe/Si bilayers, induced by 100 keV ^{40}Ar ions at room temperature was investigated. Rutherford backscattering spectroscopy (RBS), atomic force microscopy (AFM), scanning electron microscopy (SEM) and transmission electron microscopy (TEM) were applied for structural characterization. The main focus of this study was on the influence of the substrate structure on interface mixing. The influence of the substrate structure is due to the two classes of irradiated bilayers, Fe thin films deposited on crystalline or pre-amorphized Si substrates. An about 76% higher efficiency of atomic transport across the pre-amorphized Fe/a-Si interface as compared to that of Fe/c-Si bilayers was observed.

Introduction

Tailoring thin films of new materials by ion bombardment of layered structures, as an interesting alternative to physical or chemical vapor co-deposition, pulsed-laser deposition, and other thin film deposition methods, has been the subject of numerous studies. Moreover, the precise control of irradiation parameters along with the selection of suitable materials offers advantages in structuring materials on a nanometre scale, nowadays one of the most relevant technological challenges. In the past, silicide formation via ion beam mixing of metal/Si bilayers was studied in great detail (Reader *et al.*, 1993; Dhar *et al.*, 1996).

The present work outlines the influence of the structural properties of Si substrate on the intermixing processes induced by noble gas ion irradiation. The process of amorphization is important in semi-conductor processing, since Si pre-amorphization allows the formation of ultra shallow junctions.

By contrast, fundamental understanding of the ion beam-induced amorphization process in silicon is of continued interest. In this context the effect of 1.0-keV Ar^+ -ion pre-implantation of the Si substrates was studied. Such treatments may induce the amorphization of the region in proximity of the Si wafer surface, changes of the surface topography and roughness, and lowering the concentration of the contaminants, such as hydrocarbon and oxide layers at the Fe/Si interface. The results of these investigations, which were obtained with 100-keV Ar ions, revealed a pronounced enhancement of the mixing rate in the pre-amorphized samples.

Materials and methods

The Fe/Si bilayers were prepared by electron-beam evaporation using the thin film deposition facility at VINCA. The substrates used in this experiment were crystalline and pre-amorphized Si (100) wafers. For one set of the samples labelled Fe/a-Si, the Si substrates were bombarded with 1.0-keV Ar ions. The pre-amorphization process was immediately followed by deposition of Fe thin films inside of the deposition chamber. Another set of bilayers labelled Fe/c-Si was prepared by deposition of Fe thin films on a crystalline Si substrate. The Fe films were deposited on room temperature to the thicknesses of 30–32 nm, at an average deposition rate of 0.1 nm/s and a pressure of 1×10^{-4} Pa. The ion irradiations were performed with 100-keV Ar^+ ions provided by the 500-keV IONAS accelerator facility in Göttingen. All irradiations were performed at room temperature (RT). The ion fluences were in the range from 1×10^{15} to 2×10^{16} ions cm^{-2} and the implanted area was $1 \times 1 \text{ cm}^{-2}$ homogeneously covered by means of an x - y sweeping system. The RBS (Rutherford Backscattering Spectrometry) analyses were performed with a 900-keV He^{2+} ion beam provided by the

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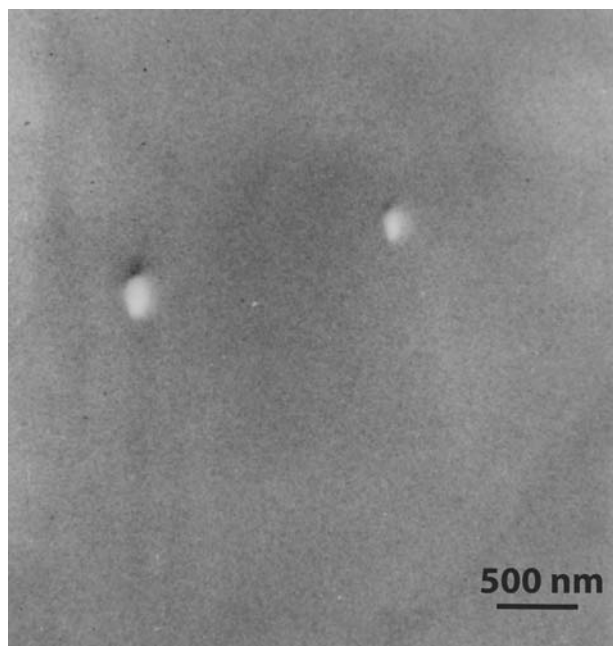


Fig. 1. SEM micrograph taken from the Si substrate bombarded with 1-keV Ar ions.

IONAS facility in Göttingen. Changes of the surface roughness of the samples before and after irradiation were analyzed by atomic force microscopy (AFM). Data were taken in ambient air and in tapping mode. Scans were $1.5\ \mu\text{m} \times 1.5\ \mu\text{m}$, $3\ \mu\text{m} \times 3\ \mu\text{m}$ and $5\ \mu\text{m} \times 5\ \mu\text{m}$. The surface roughness is represented by the root-mean-square (rms) value. The analysis of surface topography of Si substrate after the pre-amorphization process was performed by scanning electron microscopy (SEM), using a JSM-35 (JEOL) instrument (operated at 25 kV) (JEOL, Tokyo, Japan). A microstructural characterization was performed by means of high-resolution cross-sectional transmission electron microscopy (TEM), using Philips EM 400 T and CM 200 electron microscopes (Philips, Eindhoven, the Netherlands).

Results and discussion

The surface roughness (rms value) of Fe/a-Si bilayers was deduced from the AFM measurements. With $3\ \mu\text{m} \times 3\ \mu\text{m}$ scan area we obtained 1.5 nm rms roughnesses. This implies that the contribution of the surface roughness to the intermixing of the components in subsequent ion beam mixing processes is negligible.

The SEM micrograph in Fig. 1 shows the surface topography of silicon substrate after bombardment with 1.0-keV Ar ions. The Si substrate is flat and uniform, whereas the hillocks in the center correspond to the adsorbed impurities. No changes of the surface topography after pre-amorphization as compared to c-(100) Si substrate were observed.

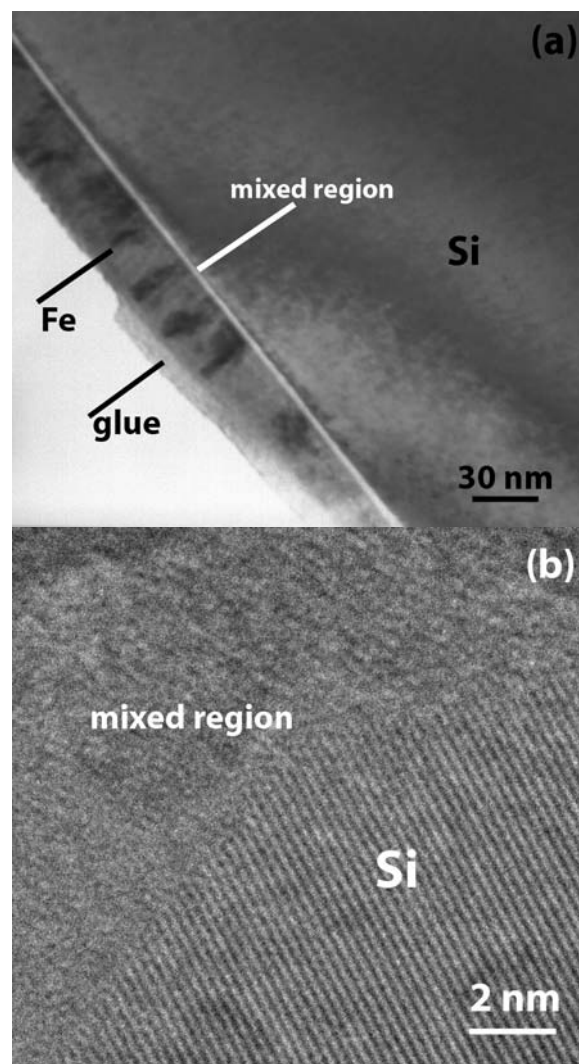


Fig. 2. TEM analysis of a Fe/a-Si sample irradiated with $1 \times 10^{16}\ \text{Ar}^+$ ions cm^{-2} : bright-field image taken from the sample aligned along [110] Si (a) and high-resolution image taken along [110] Si (b).

Figures 2(a) and (b) show cross-sectional transmission electron microscopy (x-TEM) analysis of a Fe/a-Si sample irradiated with 100-keV Ar^+ ions to $1 \times 10^{16}\ \text{cm}^{-2}$. In (a), the sample was aligned along the [110] Si zone axis. Note the high contrast between the Si substrate and the mixed region, as well as the polycrystalline structure of the top Fe layer. Here we also see a bit of glue on the top of Fe, remaining after sample preparation. The contrast arising from the substrate and the mixed region is nearly equalized, showing that the top Fe layer and the mixed region are closely bound to the Si substrate. A detailed micro diffraction (MD) analysis for various sample tilts, covering an area of approximately 35 nm in diameter centred on the intermixed region indicated that the bright-contrast region is amorphous. This leads to the conclusion that the intermixed region has an amorphous

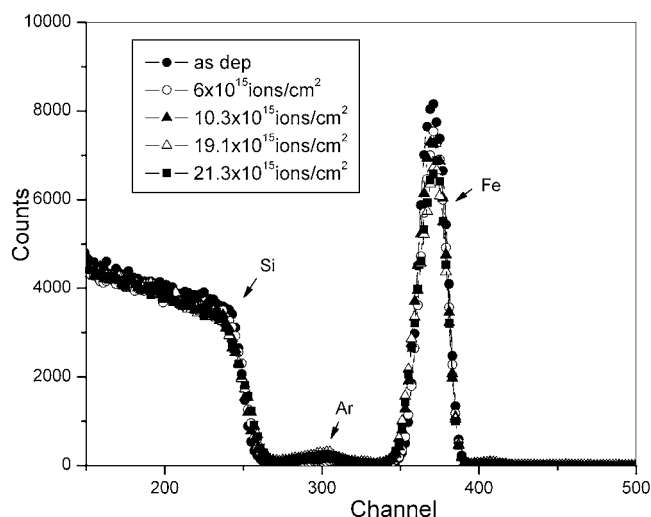


Fig. 3. RBS spectra taken before and after 100-keV Ar^+ ion irradiations of Fe/a-Si samples.

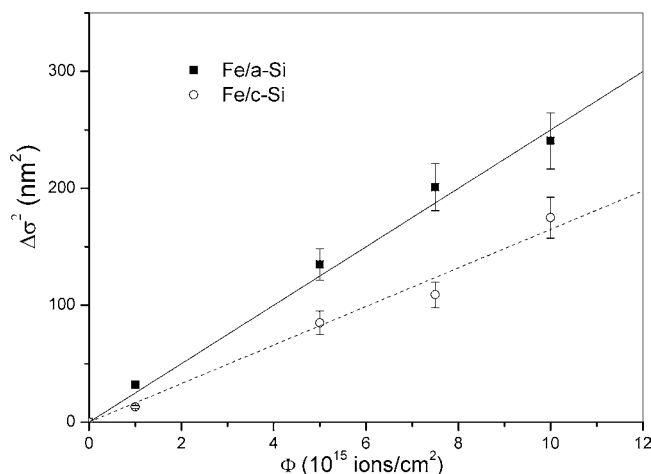


Fig. 4. Interface broadening variance $\Delta\sigma^2(\Phi)$ versus ion fluence Φ for Fe/c-Si and Fe/a-Si bilayers irradiated with 100-keV Ar^+ ions.

structure. Furthermore, we did TEM analyses using higher magnifications and different sample tilts, which showed no diffraction contrasts within the mixed layer, indicating that this region is fully amorphous. An HRTEM image, taken along $[110]$ Si and shown in Fig. 2(b), exhibits lattice $\{111\}$ Si fringes from the substrate (0.313 nm spacing), and again a different crystalline structure of the mixed region. The analysis of different areas of the sample showed a similar uniform contrast of the intermixed region.

The RBS spectra presented in Fig. 3 were taken from an as-deposited 30-nm Fe/a-Si sample and from samples irradiated with 100-keV Ar^+ ions at RT. Irradiation fluences were in

the range of $0.5\text{--}2 \times 10^{16}$ ions cm^{-2} . The spectra clearly show that ion irradiation induces intermixing of the bilayer components at the Fe/a-Si interface. The Ar yield is well separated on the RBS spectra and is located at the metal/silicon interface. On the basis of the Si and Fe depth profiles obtained from RBS, the changes of the variance σ^2 of the Fe and Si concentration profiles were deduced as a function of the ion fluence Φ providing the mixing rates $k \equiv \Delta\sigma^2/\Phi$. Figure 4 illustrates the $\Delta\sigma^2$ -versus- Φ plot corresponding to the Fe/c-Si and Fe/a-Si bilayers. In both cases, the interface variance $\Delta\sigma^2$ varies linearly with the ion fluence Φ . It can be clearly seen that the mixing rate ($k \equiv \Delta\sigma^2/\Phi$) is higher for the Fe/a-Si bilayers. Since all the other experimental parameters were kept identical for both the Fe/a-Si and Fe/c-Si bilayers, we argue that amorphization of the Si wafer is the major reason for intermixing enhancement. This can be related to the smaller cooling rate in a-Si ($0.3 \times 10^9 \text{ K s}^{-1}$) than that in c-Si ($2.5 \times 10^9 \text{ K s}^{-1}$) (Schwickert *et al.*, 2003). In the case of interface mixing this may lead to a longer diffusion time and to a higher mixing rate. Consequently, we introduced the ratio of mixing rates for amorphized and crystalline Si, $R_{a/c} \equiv [\Delta\sigma^2/\Phi]_a/[\Delta\sigma^2/\Phi]_c$. For the measured values of $R_{a/c}$, we arrived at the weighted mean value of $R_{a/c} = 1.76(15)$ (Bibić *et al.*, 2007).

Conclusions

The present careful mixing experiments on Fe/Si bilayers with 100-keV Ar ions have given clear evidence of enhanced mixing rates, whenever the Si wafers were pre-amorphized with 1.0-keV Ar ions before deposition of the 30-nm Fe top layer. This enhancement by 76% relative to crystalline Si wafers may be related to a longer diffusion time in amorphous silicon.

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